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# Structure of $\mathrm{Na}_{2} \mathbf{S O}_{4}(\mathbf{I})$ at 693 K 

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#### Abstract

The crystal structure of $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ at 693 K was investigated by X-ray diffraction and refined to an $R$ value of 0.049 for 63 observed independent reflections. $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ has the space group $P 6_{3} / m m c$ with $a=5.444(2), c=7 \cdot 347$ (6) $\AA, \quad V=188 \cdot 6(2) \AA^{3}$ and $Z=2 . M_{r}=142 \cdot 04, D_{x}=2 \cdot 501 \mathrm{Mg} \mathrm{m}^{-3}, F(000)=140$ and $\mu(\mathrm{MoK} \alpha)=0.9466 \mathrm{~mm}^{-1}$. The $\mathrm{SO}_{4}$ tetrahedron mainly adopts the apex model orientation with one of the apices pointing statistically up and down the $c$ axis. Further, the apical O atom is not exactly on the threefold axis passing through the central S atom, but deviates slightly from it. A small fraction of $\mathrm{SO}_{4}$ groups take the orientation of the edge model at 693 K.


## Introduction

Five modifications designated as phases (I) to (V) have been reported for $\mathrm{Na}_{2} \mathrm{SO}_{4}$ crystals to date (Kracek \& Gibson, 1930; Kracek \& Ksanda, 1930; Saito, Kobayashi \& Maruyama, 1982; Eysel, Höfer, Keester \& Hahn, 1985). Among these modifications, $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{IV})$ is supposed to be metastable and the phase relation has not yet been well established. Precise structure determinations of the room-temperature
phase $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{~V})$ and of the intermediate phase $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (III) were carried out by Nord (1973) and Mehrotra (1981), respectively. Since the stable temperature range of $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (II) is narrow, the structural details of $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (II) are unknown. Several structure models have been proposed for the high-temperature phase $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$. The arrangement of $\mathrm{Na}^{+}$and $\mathrm{SO}_{4}^{2-}$ ions is the same in all these models except for the orientation of $\mathrm{SO}_{4}^{2-}$ groups.
Recently, Eysel et al. (1985) carried out a detailed structure investigation of yttrium-stabilized $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ at room temperature, and revealed that the structure agrees well with the model proposed by Höfer (1979), where one of tetrahedral edges of the $\mathrm{SO}_{4}^{2-}$ group is parallel to the $c$ axis of the hexagonal lattice and statistically the anion takes three crystallographically equivalent orientations in equal probability around the threefold rotation axis. However, the mode of the statistical distribution may change with temperature as pointed out by Eysel et al. (1985). Therefore, it is of interest to obtain the structural details of $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ at high temperatures where the modification is stable.
Amirthalingam, Karkhanavala \& Rao (1977) reported that crystals of $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (III), which show topotactic transformation to $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$, remain
transparent when they are heated to transform into $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ and cooled again to $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (III). A series of structural investigations were carried out on $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (I) and $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (III) by making use of crystals of $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (III). In the present paper, the crystal structure of $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ at 693 K is studied by the singlecrystal X-ray diffraction method. The thermal change of anharmonic vibrations of O atoms in $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (III) will be reported in a forthcoming paper.

## Experimental

$\mathrm{Na}_{2} \mathrm{SO}_{4}$ from Wako Pure Chemical Industries, Japan, was dissolved in distilled water. The aqueous solution was quickly evaporated at about 333 K , giving needles of $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (III) together with bipyramidal crystals of $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{~V})$.

The crystal used for intensity measurements had dimensions $0.04 \times 0.07 \times 0.41 \mathrm{~mm}$. It was set on the top of a fused-silica capillary (diameter 0.3 mm ) with adhesive Sauereisen cement. A Pt-Pt13\% Rh thermocouple (diameter 0.1 mm ) was placed 0.3 mm below the sample through the capillary for temperature measurement. A heating apparatus for a four-circle diffractometer developed by Ishizawa \& Kato (1983) was used to heat the specimen. The temperature stability was within $\pm 3 \mathrm{~K}$ up to 1273 K .

Intensities were measured at 693 K on a four-circle diffractometer (Rigaku AFC-5UD) with Mo $K \alpha$ radiation $(\lambda=0.71069 \AA)$ monochromated with pyrolytic graphite. The $\omega$-scan mode was employed with a scanning speed of $2^{\circ} \mathrm{min}^{-1}$ in $\omega$. Because of the diffuseness of reflection spots, large scan widths were adopted according to the formula $(4 \cdot 5+\tan \theta)^{\circ}$. The intensities were corrected for Lorentz-polarization factors and for absorption effects, assuming a rectangular shape for the crystal. The absorption correction factors were in the range $1 \cdot 012-1 \cdot 041$.

A total of 106 reflections which satisfied $\left|F_{o}\right|>$ $3 \sigma\left(\left|F_{o}\right|\right)$ were obtained within the range $2 \theta<50^{\circ}$, $0 \leq h \leq 5,0 \leq k \leq 5$ and $-8 \leq l \leq 0$, and used for the subsequent refinements ( 176 reflections were measured, of which 70 were unobserved; there were 63 unique observed reflections). Here, $\sigma\left(\left|F_{o}\right|\right)$ is the estimated standard deviation due to counting statistics. The lattice parameters were obtained from 43 reflections in the range $14<2 \theta<30^{\circ}$ by the leastsquares procedure. Dimensions of the hexagonal unit cell are $a=5.444$ (2) and $c=7.347$ (6) $\AA$ at 693 K . The observed systematic absences of hhl for $l$ odd restricted the possible space groups to $P 6_{3} / \mathrm{mmc}$, $P \overline{6} 2 c$ and $P 6_{3} m c$.

## Structure refinement

The three-dimensional Patterson map showed Na at $(0,0,0),\left(0,0, \frac{1}{2}\right),\left(\frac{1}{3}, \frac{2}{3}, \frac{3}{4}\right)$ and $\left(\frac{2}{3}, \frac{1}{3}, \frac{1}{4}\right)$, and $S$ at $\left(\frac{1}{3}, \frac{2}{3}, \frac{1}{4}\right)$ and $\left(\frac{2}{3}, \frac{1}{3}, \frac{3}{4}\right)$. These atomic positions are idencical to

Table 1. Atomic parameters and anisotropic temperature factors $\left(\AA^{2}\right)$ of $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ at 693 K

Standard deviations in parentheses refer to the last significant figures. The anisotropic temperature factors are expressed in the form: $\exp \left[-2 \pi^{2}\left(U_{11} h^{2} a^{* 2}+U_{22} k^{2} b^{* 2}+U_{33} l^{2} c^{* 2}+2 U_{12} h k a^{*} b^{*}+2 U_{13} h l a^{*} c^{*}\right.\right.$ $\left.\left.+2 U_{23} k l b^{*} c^{*}\right)\right]$.

|  | $x$ | $y$ | $z$ | $U_{11} \dagger$ | $U_{22}$ | $U_{33}$ | $U_{13}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Na}(1)$ | 0.0 | 0.0 | 0.0 | $0.083(3)$ | $U_{11}$ | $0.090(5)$ | 0.0 |
| $\mathrm{Na}(2)$ | $\frac{1}{3}$ | $\frac{2}{3}$ | $\frac{3}{4}$ | $0.082(4)$ | $U_{11}$ | $0.166(8)$ | 0.0 |
| S | 0.3 | $\frac{2}{3}$ | $\frac{1}{4}$ | $0.048(2)$ | $U_{11}$ | $0.069(3)$ | 0.0 |
| $\mathrm{O}(1)$ | $0.271(5)$ | $2 x$ | $0.450(3)$ | $0.25(13)$ | $0.14(11)$ | $0.08(2)$ | $0.03(3)$ |
| $\mathrm{O}(2)$ | $0.193(1)$ | $2 x$ | $0.226(6)$ | $0.12(1)$ | $0.08(2)$ | $0.31(2)$ | $-0.06(2)$ |
|  |  | $\dagger 2 U_{12}=U_{11}$ | and $U_{23}=2 U_{13}$. |  |  |  |  |

those in the structure models previously reported. The Fourier and difference Fourier maps synthesized with phases derived from Na and S atoms gave three peaks on the plane $z=\frac{1}{4}$ around the $S$ atom at $\left(\frac{1}{3}, \frac{2}{3}, \frac{1}{4}\right)$, which were supposed to be O atoms. Besides them, two peaks were observed above and below the $S$ atom, having heights a half of those for the peaks on the $z=\frac{1}{4}$ plane. Initially, structure refinement was tried by allotting O atoms to the three peaks on $z=\frac{1}{4}$ and to the peak above the S atom, assuming the space group $\mathrm{Pb}_{3} m c$. However, the peak below S did not disappear on the difference Fourier map after this refinement, suggesting the true space group $\mathrm{P6}_{3} / \mathrm{mmc}$ with statistical distrıbution of the $\mathrm{SO}_{4}$ tetrahedron between the two orientations.

The structure was therefore refined with the fullmatrix least-squares program LINEX (Becker \& Coppens, 1974a,b, 1975) on the basis of symmetry $P 6_{3} / \mathrm{mmc}$, by taking into account the orientational disorder of the $\mathrm{SO}_{4}$ tetrahedron over the two possible orientations. The refinement with anisotropic temperature factors gave a $w R$ value of $0 \cdot 054$. At this stage, Fourier and difference Fourier maps were synthesized. These maps revealed that the $O(1)$ atom has a triradiated electron cloud and $\mathrm{O}(2)$ an extraordinarily elongated electron cloud along the $c$ axis. The final refinement was carried out by splitting the $\mathrm{O}(1)$ atom further into three located at positions deviating slightly from the threefold axis, giving $R$ and $w R$ values of 0.056 and 0.051 for the 106 observed reflections, and 0.049 and 0.044 for the unique reflections. The atomic scattering factors for neutral atoms were taken from International Tables for X-ray Crystallography (1974). The function $\Sigma w\left(\left|F_{o}\right|-\left|F_{c}\right|\right)^{2}$ was minimized, assuming $w=1$ for all reflections. The final positional and thermal parameters are given in Table 1.* Maximum $\Delta / \sigma=0.032$. The final residual densities are between 0.23 and $-0.53 \mathrm{e}_{\AA^{-3}}$. Bond distances and angles are listed in Table 2 after correc-

[^0]Table 2. Interatomic distances ( $\AA$ ) and angles $\left({ }^{\circ}\right)$ in $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ at 693 K

| $\mathrm{S}^{\mathrm{i}}$-O( $\mathrm{I}^{\text {i }}$ ) | $(\times 1)$ | 1.58 (2) | 1.70 (2)* |
| :---: | :---: | :---: | :---: |
| $\mathrm{S}^{\mathrm{i}} \mathrm{O} \mathrm{O}\left(2^{\mathrm{i}}\right.$ ) | ( $\times 3$ ) | 1.33 (1) | 1.47 (1)* |
|  | average | 1.39 | 1.53 |
| $\mathrm{O}\left(1^{1}\right)-\mathrm{O}\left(2^{2}\right)$ | $(\times 2)$ | $2 \cdot 13$ (4) |  |
| $\mathrm{O}\left(1^{\text {i }}\right.$ ) $-\mathrm{O}\left(2^{\text {ii }}\right.$ ) | ( $\times 1$ ) | 2.36 (4) |  |
| $\mathrm{O}\left(2^{i}\right)-\mathrm{O}\left(2^{2}\right)$ | $(\times 2)$ | 2.31 (1) |  |
| $\mathrm{O}\left(2^{\text {i }}\right.$ ) $-\mathrm{O}\left(2^{\text {ii }}\right.$ ) | ( $\times 1$ ) | 2.28 (1) |  |
|  | average | 2.25 |  |
| $\mathrm{O}\left(1^{1}\right)-\mathrm{S}^{\mathrm{i}}-\mathrm{O}\left(2^{\text {ii }}\right.$ ) | ( $\times 1$ ) | 108.0 (1.0) |  |
| $\mathrm{O}\left(1^{i}\right)-\mathrm{S}^{\mathrm{i}}-\mathrm{O}\left(2^{\text {l }}\right.$ ) | $(\times 2)$ | 93.4(1.1) |  |
| $\mathrm{O}\left(2^{\text {i }}\right.$ ) $-\mathrm{S}^{\mathrm{i}}-\mathrm{O}\left(2^{\text {ii }}\right.$ ) | ( $\times 1$ ) | 118.2 (0.8) |  |
| $\mathrm{O}\left(2^{i}\right)-\mathrm{S}^{\mathrm{i}}-\mathrm{O}\left(2^{\text {v }}\right.$ ) | $(\times 2)$ | 120.6 (0.8) |  |
| $\mathrm{Na}\left(1^{i}\right)-\mathrm{O}\left(1^{\text {iv }}\right.$ ) | ( $\times 1$ ) | 2.58 (2) |  |
| $\mathrm{Na}\left(1^{1}\right)-\mathrm{O}\left(1^{\text {v }}\right.$ ) | ( $\times 2$ ) | 3.49 (3) |  |
| $\mathrm{Na}\left(1^{i}\right)-\mathrm{O}\left(2^{\text {i }}\right.$ ) | ( $\times 3$ ) | 2.47 (3) |  |
| $\mathrm{Na}\left(\mathrm{I}^{\text {i }}\right.$ ) $-\mathrm{O}\left(2^{\text {iv }}\right.$ ) | $(\times 3)$ | 2.72 (4) |  |
|  | average | 2.79 |  |
| $\mathrm{Na}\left(2^{\mathrm{i}}\right)-\mathrm{O}\left(1^{\text {i }}\right.$ ) | $(\times 1)$ | 2.28 (2) |  |
| $\mathrm{Na}\left(2^{\mathrm{i}}\right)-\mathrm{O}\left(1^{\text {iiii }}\right)$ | ( $\times 3$ ) | 3.25 (3) |  |
| $\mathrm{Na}\left(2^{\text {i }}\right.$ ) $-\mathrm{O}\left(2^{\text {iii }}\right)$ | ( $\times 6$ ) | 2.74 (1) |  |
| $\mathrm{Na}\left(2^{\text {i }}\right.$ ) $-0\left(2^{\text {iv }}\right)$ | ( $\times 3$ ) | 3.74 (4) |  |
|  | average | 3.05 |  |

Symmetry code: (i) $x, y, z$; (ii) $1-y, 1+x-y, z$; (iii) $1-x, 1-y, 1-z$; (iv) $x, y, \frac{1}{2}-z$; (v) $y-x, 1-x, \frac{1}{2}-z$.

* Corrected values according to the riding-atom model (Busing \& Levy, 1964).
tion for thermal vibrations according to the ridingmotion model (Busing \& Levy, 1964).


## Discussion

Fig. 1 shows the cell relation between $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ and $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (III) together with the atomic arrangements in $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (III). The cell dimensions of $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (III) are $a=5.607, b=8.955(=\sqrt{3} \times 5 \cdot 170), c=6.967 \AA$ at 505 K (Mehrotra, 1981), while those of $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ are $a=5 \cdot 408, c=7 \cdot 216 \AA$ at 533 K (Mehrotra, 1973) and $a=5 \cdot 444, c=7.347 \AA$ at 693 K (this work). With increase in temperature from 505 to 533 K through the phase-transition point, the cell dimensions decrease by $3.5 \%$ along the $a$ axis and increase by $4 \cdot 6 \%$ and $3 \cdot 6 \%$ along the $b$ and $c$ axes, respectively. Above the transition point, both $a$ and $c$ increase with temperature ( $0.7 \%$ and $1.8 \%$ from 533 to 693 K along $a$ and $c$, respectively).

Since the number of observed (independent) reflections is only about four times the number of parameters to be refined, the values obtained have large standard deviations. As a consequence, an unreasonably large distortion is observed in the shape of the $\mathrm{SO}_{4}$ tetrahedron. However, several important features of the structure can be deduced from the final Fourier maps shown in Fig. 2. The maps suggest the following facts concerning the orientation of the $\mathrm{SO}_{4}$ tetrahedron: (a) the $\mathrm{SO}_{4}$ tetrahedron has one of its faces approximately parallel to (001); (b) the apex opposite to the face points up and down with equal probability; (c) the triradiated shape of the $\mathrm{O}(1)$ electron cloud indicates that $\mathrm{O}(1)$ is further split into three within
the (001) plane; (d) with a much smaller probability, the $\mathrm{SO}_{4}$ tetrahedron seems to take the orientation where one of its edges is parallel to the $c$ axis. The orientations of (a) and (d) correspond to the apex and edge models, respectively, proposed by Arnold, Kurtz, Richter-Zinnius, Bethke \& Heger (1981) for the high-temperature form of $\mathrm{K}_{2} \mathrm{SO}_{4}$.

Refinement with the edge model gave a final $w R$ value of 0.054 with anisotropic temperature factors. The positions of O atoms obtained by the refinements are indicated by the marks $X$ in Figs. 2(b) and 2(c). The marks are located on the tails of $O(2)$ peaks extending along the $\mathbf{c}$ direction as seen in Fig. 2(c). Therefore, the apex model should be the main arrangement in $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ at 693 K , and a smaller contribution is afforded by the edge model. A similar feature was observed in the high-temperature form of $\mathrm{K}_{2} \mathrm{SO}_{4}$ (Miyake, Morikawa \& Iwai, 1980). It is noteworthy that the signs of the $F$ values are


Fig. 1. The unit-cell relation between $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (III) and $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ with the atomic arrangements in $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (III) projected along (a) $c$ and (b) $a$ axes. The solid and dotted lines indicate unit cells of the phases (III) and (I), respectively.

(c)

Fig. 2. Fourier maps of $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ at 693 K through the planes $(a) z=0 \cdot 25,(b) z=0.05$ and (c) $y=2 x$. O -atom positions in the edge and apex models are marked with crosses and dots, respectively.
independent of the structure model when it is restricted to the edge and apex types mentioned above, as already pointed out by Arnold et al. (1981) for the high-temperature form of $\mathrm{K}_{2} \mathrm{SO}_{4}$.

According to the structural investigation of yttriumstabilized $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ by Eysel et al. (1985), the $\mathrm{SO}_{4}$ group mainly takes the orientation of the edge model at room temperature. However, they predicted that the apex model gains more importance at higher temperatures. The present results accord well with their prediction.
The Na atoms in $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{~V})$ and $\mathrm{Na}_{2} \mathrm{SO}_{4}$ (III) form $\mathrm{NaO}_{6}$ coordination octahedra with average $\mathrm{Na}-\mathrm{O}$ distances of $2.431 \AA$ (Nord, 1973) and $2.457 \AA$ (Mehrotra, 1981), respectively. On average, the Na atoms in $\mathrm{Na}_{2} \mathrm{SO}_{4}(\mathrm{I})$ are surrounded by seven O atoms, with mean $\mathrm{Na}(1)-\mathrm{O}$ and $\mathrm{Na}(2)-\mathrm{O}$ distances of 2.59 and $2 \cdot 67 \AA$, respectively. The $\mathrm{Na}(1)$ atom has six $\mathrm{O}(2)$ atoms arranged octahedrally around it. There are six $\mathrm{O}(1)$ positions near $\mathrm{Na}(1)(2.58 \AA)$, but each has a population of $\frac{1}{6}$. The $\mathrm{Na}(2)$ atom has-six $\mathrm{O}(2)$ atoms roughly on the plane through $\mathrm{Na}(2)$ and parallel to (001). The $\mathrm{O}(1)$ positions, approximately on the line through $\mathrm{Na}(2)$ and parallel to the $c$ axis, are at a distance of $2 \cdot 28 \AA$. Since this distance is too short for an $\mathrm{Na}-\mathrm{O}$ bond, the structure should have a static deviation of the $\mathrm{Na}(2)$ atom from the mean position along the $\mathbf{c}$ direction. Actually, the electron density
map shows elongation of the $\mathrm{Na}(2)$ electron cloud along the $\mathbf{c}$ direction.

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[^0]:    * A list of observed and calculated structure factors has been deposited with the British Library Document Supply Centre as Supplementary Publication No. SUP 43327 ( 2 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

